



Ferroelectric AgNa(NO₂)₂ crystals as novel highly efficient nonlinear optical material: Phase matched second harmonic generation driven by a spontaneous and electric field induced polarizations

Bouchta Sahraoui, Robert Czaplicki, A. Klopperpieper, A.-S. Andrushchak,
A.-V. Kityk

► To cite this version:

Bouchta Sahraoui, Robert Czaplicki, A. Klopperpieper, A.-S. Andrushchak, A.-V. Kityk. Ferroelectric AgNa(NO₂)₂ crystals as novel highly efficient nonlinear optical material: Phase matched second harmonic generation driven by a spontaneous and electric field induced polarizations. *Journal of Applied Physics*, 2010, 107 (11), Non spécifié. 10.1063/1.3415545 . hal-03343401

HAL Id: hal-03343401

<https://univ-angers.hal.science/hal-03343401>

Submitted on 14 Sep 2021

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.



Ferroelectric AgNa (NO 2) 2 crystals as novel highly efficient nonlinear optical material: Phase matched second harmonic generation driven by a spontaneous and electric field induced polarizations

B. Sahraoui, R. Czaplicki, A. Klöpperpieper, A. S. Andrushchak, and A. V. Kityk

Citation: *Journal of Applied Physics* **107**, 113526 (2010); doi: 10.1063/1.3415545

View online: <http://dx.doi.org/10.1063/1.3415545>

View Table of Contents: <http://scitation.aip.org/content/aip/journal/jap/107/11?ver=pdfcov>

Published by the AIP Publishing

Articles you may be interested in

[Multi-direction high-efficiency second harmonic generation in ellipse structure nonlinear photonic crystals](#)
Appl. Phys. Lett. **105**, 151106 (2014); 10.1063/1.4898187

[Large second-order optical nonlinearity in a ferroelectric molecular crystal of croconic acid with strong intermolecular hydrogen bonds](#)

Appl. Phys. Lett. **102**, 162901 (2013); 10.1063/1.4802727

[Spontaneous and electric field induced quadratic optical nonlinearity in ferroelectric crystals AgNa \(NO 2 \) 2](#)
Appl. Phys. Lett. **96**, 061911 (2010); 10.1063/1.3315941

[Broadband quasi-phase-matched second-harmonic generation in a nonlinear photonic crystal](#)
J. Appl. Phys. **100**, 023120 (2006); 10.1063/1.2213150

[Electric field induced second harmonic generation with and without fringes](#)
Rev. Sci. Instrum. **71**, 3490 (2000); 10.1063/1.1287629

An advertisement for AIP Applied Physics Reviews. It features a blue and orange design with molecular models. On the left, there's a thumbnail of the journal cover showing a grid and some text. The main title "NEW Special Topic Sections" is in large white letters. Below it, "NOW ONLINE" is in yellow, followed by the title "Lithium Niobate Properties and Applications: Reviews of Emerging Trends". The AIP logo is at the bottom right.

Ferroelectric AgNa(No₂)₂ crystals as novel highly efficient nonlinear optical material: Phase matched second harmonic generation driven by a spontaneous and electric field induced polarizations

B. Sahraoui,¹ R. Czaplicki,¹ A. Klöpperpieper,² A. S. Andrushchak,³ and A. V. Kityk^{4,a)}

¹Institut des Sciences et Technologies Moléculaires d'Angers, MOLTECH ANJOU-UMR CNRS 6200,

Université d'Angers, 2 bd Lavoisier, 49045 Angers Cedex 2, France

²Technische Physik, Universität des Saarlandes, 66041 Saarbrücken, Germany

³Lviv Polytechnic National University, 12 S. Bandera Str., 79013 Lviv, Ukraine

⁴Faculty of Electrical Engineering, Czestochowa University of Technology, Al. Armii Krajowej 17, 42-200 Czestochowa, Poland

(Received 7 February 2010; accepted 3 April 2010; published online 7 June 2010)

Paper reports the second harmonic generation (SHG) in ferroelectric AgNa(No₂)₂ crystals being driven by the spontaneous and electric field induced polarizations. Obtained results are interpreted within the phenomenological theory which considers the free energy describing the interaction between the spontaneous or electric field induced polarizations and spatially inhomogeneous electric polarizations resulted from propagating optical waves. Relatively high magnitudes of the effective second order nonlinear optical (NLO) susceptibilities in these crystals are combined with several phase matching geometries which allows to consider them as high-performance materials for potential NLO applications, such as parametric generation and amplification, frequency doubling, or other applications that require high-efficient frequency conversion. In addition, an anomalously large response of NLO susceptibilities with respect to an applied electric field has been found in the vicinity of the Curie point. This may also have a number of applications, especially in those devices where an efficient tunable control of SHG intensity is demanded. © 2010 American Institute of Physics. [doi:10.1063/1.3415545]

I. INTRODUCTION

Ferroelectric crystals are considered as very promising materials for a number of optoelectronic applications. Most of them are based on parametrical or nonlinear optical (NLO) phenomena such as electro-optic effect, second harmonic generation (SHG), parametric down conversion, parametric amplification or generation, etc., being widely used for high-efficient light modulation or its frequency conversion.^{1–3} The SHG, known also as frequency doubling, represents the phenomenon tightly coupled with nonlinear second order optical susceptibilities $\chi_{ijk}^{(2)}$ which are the components of third order polar tensor. Most of the ferroelectric materials are centrosymmetric in paraelectric phase, thus all the NLO susceptibilities $\chi_{ijk}^{(2)}=0$, i.e., the SHG is forbidden by the symmetry. Upon cooling a phase transition (PT) to the ferroelectric state breaks the symmetry leading to a dramatic change in a SHG efficiency. A spontaneous polarization P_s plays, in such transformation, a role of symmetry breaking parameter. It describes quantitatively structural changes below the Curie point T_c which result to a lowering of the crystal symmetry. Accordingly, the behavior of NLO susceptibilities $\chi_{ijk}^{(2)}$ in the ferroelectric phase correlate with the P_s thereby the SHG may be considered here as an effect driven by the spontaneous polarization and, in analogy to the electro-optical terminology,⁴ one may call it as *spontaneous SHG*.⁵

Alternatively, the inversion center of the parent structure may be lost under an electric field applied to the crystal. This also results in SHG. Its magnitude is essentially related to the electric polarization of the media, thus it is proportional to the dielectric constant. For this reason the *field induced SHG* is usually a weak effect as long as the classical linear dielectrics are considered. The situation changes cardinally if one deals with the proper ferroelectrics. Their dielectric constant along the polar axis behaves anomalously in the vicinity of the Curie point T_c .⁶ Accordingly, the SHG, driven by an applied electric field may rise here even up to several orders of magnitude. This is of a considerable interest in a light of a large number of applications. In particular, by choosing a temperature range close to T_c one may reach a considerable modulation efficiency and/or substantially reduce an operating voltage of corresponding optoelectronic devices. Other types of applications require tunable intensity of SHG signal which in the case of an optical cell made of a ferroelectric crystal material can be easily adjusted either by a temperature change and/or by an electric field tuning.

In this work, we report the phase matched SHG in ferroelectric AgNa(No₂)₂ being driven by spontaneous and electric field induced polarizations. Actual paper represents considerable extensions of the data published recently in the letter article.⁵ The AgNa(No₂)₂ crystals exhibit the PT at $T_c \approx 38$ °C (Refs. 7 and 8) from a paraelectric phase (space group D_{2h}²⁴) to the proper ferroelectric phase (space group C_{2v}¹⁹). The dielectric,^{7–9} specific-heat,¹⁰ and elastic measurements¹¹ have revealed that the PT is of the first order, however very close to the second order one. For this

^{a)}Author to whom correspondence should be addressed. Electronic mail: andriy.kityk@uvivie.ac.at

reason the anomalies of dielectric and thermal properties can be well explained by a Landau theory taking into account the near tricritical character of this PT. As a consequence the leading term in the free energy expansion is of the sixth order in the polarization P . Ferroelectricity in $\text{AgNa}(\text{NO}_2)_2$ appears due to an ordering of the NO_2^- dipoles¹² which at room temperature give quite large spontaneous polarization (about $8 \mu\text{C}/\text{cm}^2$) being oriented along [010] crystallographic direction.⁸ The uniaxial ferroelectric $\text{AgNa}(\text{NO}_2)_2$ constitutes an exception in the sense of its polarization dynamics which is of Debye-type and appears in the kilohertz region.^{9,13,14}

II. BRIEF PHENOMENOLOGICAL ANALYSIS OF THE QUADRATIC OPTICAL NONLINEARITY IN $\text{AGNA}(\text{NO}_2)_2$

In this section, we present a brief analysis of the second order optical nonlinearity in ferroelectric $\text{AgNa}(\text{NO}_2)_2$. It is based on the phenomenological theory and allows to understand a relation between the NLO susceptibilities $\chi_{ijk}^{(2)}$ and the basic macroscopic characteristics of this ferroelectric material, i.e., its spontaneous polarization P_s and/or dielectric susceptibility. We start from the general characterization of the ferroelectric PT in $\text{AgNa}(\text{NO}_2)_2$ crystals. Following the actual symmetry transformation ($D_{2h} \rightarrow C_{2v}$) the Landau free energy may be written as:¹¹

$$F_P = \frac{1}{2}A(T - T_0)P^2 + \frac{1}{4}BP^4 + \frac{1}{6}CP^6 - PE, \quad (1)$$

where P is the Y -component of the polarization being the order parameter of this model, A , B , and C are the free energy expansion coefficients, T_0 is the Curie–Weiss temperature, and E is the electric field. Specific of the paraelectric-to-ferroelectric PT in $\text{AgNa}(\text{NO}_2)_2$ causes several requirements in regard to the expansion coefficients. We indeed deal with a weakly first order transition being very close to the tricritical point. Such formulation is equivalent to the case when the coefficient B is represented by small negative value whereas the two other coefficients A and C are essentially positive. Considering the case when $E=0$ the actual PT temperature T_c and the equilibrium value of the spontaneous polarization P_s may be obtained in usual way by solving simultaneously a set of the two following equations:

$$F_P = 0, \quad \partial F_P / \partial P = 0, \quad (2)$$

which yields:

$$T_c = T_0 + 3B^2/16AC \quad (3)$$

and

$$P_s = 0 \quad (T > T_c),$$

$$P_s^2 = (-B + \sqrt{B^2 - 4CA(T - T_0)})/2C \quad (T \leq T_c). \quad (4)$$

Figure 1(a) shows schematically the T -dependence of P_s ($E=0$, see solid line of blue color) in the vicinity of the Curie point T_c . If the electric field is applied ($E \neq 0$) the equilibrium polarization $P^+ = P_s + P_{\text{ind}}$ (P_{ind} is the field induced polarization) may be obtained by solving the electric equation

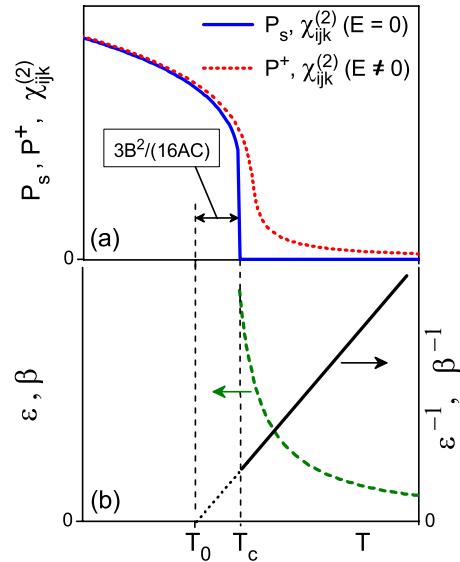


FIG. 1. (Color online) Temperature behavior of P_s , P^+ , and $\chi_{ijk}^{(2)}$ (a); ϵ , β , ϵ^{-1} , and β^{-1} vs T (b) near the paraelectric-to-ferroelectric PT (schematic presentation).

of state ($\partial F_P / \partial P = -E$). This leads to the dependence P^+ versus T as qualitatively shown by the dashed red line in Fig. 1(a). It should be emphasized that the crystal symmetry is different on both sides of the PT point only when $E=0$. For $\text{AgNa}(\text{NO}_2)_2$ it is centrosymmetric (D_{2h}) above T_c and non-centrosymmetric (C_{2v}) below it. If $E \neq 0$ the symmetry is identical on both sides of the Curie point (C_{2v}) thus the PT loses its classical meaning because there is no break of the symmetry anymore here. Nevertheless, even at $E \neq 0$ most of physical properties keep its anomalous behavior in the vicinity of Curie point. One of such characteristics, being important for the further analysis of NLO properties, is the static dielectric susceptibility $\chi = \epsilon_0^{-1} \partial P / \partial E$. Its inverse magnitude may be obtained by differentiating the electric equation of state

$$(\epsilon_0 \chi)^{-1} = \partial E / \partial P = A(T - T_0) + 3BP^2 + 5CP^4. \quad (5)$$

Above T_c and at small electric field ($E \rightarrow 0$) the induced polarization $P_{\text{ind}} \equiv P^+$ can be ignored in Eq. (5) (since $P^+ \rightarrow 0$), accordingly, the static dielectric constant $\epsilon = 1 + \chi \approx C_0/(T - T_0)$, where $C_0 = (A\epsilon_0)^{-1}$ is the Curie–Weiss constant. ϵ and ϵ^{-1} versus T are shown schematically in Fig. 1(b). Obviously, the strength of dielectric anomaly near the Curie point T_c is a matter of its closeness to the tricritical point at which the difference $(T_c - T_0) \rightarrow 0$ and $\epsilon(T_c) \rightarrow \infty$. $\text{AgNa}(\text{NO}_2)_2$ in fact constitutes an example of the crystal with a weakly first order paraelectric-to-ferroelectric PT. For this reason the static dielectric constant indeed rises in the vicinity T_c on about of the two orders of magnitude.^{7–9}

To interpret the NLO properties we introduce into the free energy the electric polarizations induced by a propagating electromagnetic light waves. The approach offered here is in agreement with the Muller's formalism developed in Ref. 15. However, it is more general since we deal with the inhomogeneous free energy instead of the free energy in a local approximation as presented in Ref. 15. In some respect the forthcoming approach is based on principles similar to

those used in the description of incommensurate phases (see, e.g., Refs. 16–19). The part of the free energy density which is responsible for the SHG in ferroelectric phase of $\text{AgNa}(\text{NO}_2)_2$ may be written as:

$$f_{\text{int}}(r,t) = -\alpha_{2jkl}^{0,\omega,\omega,2\omega}[P_2(0)P_j(\omega,k_1)P_k(\omega,k_2)P_l^*(2\omega,k_3) + c.c.] + \frac{1}{2\varepsilon_0\chi_{ii}^\Omega}P_i(\Omega,K)P_i^*(\Omega,K), \quad (6)$$

where ε_0 is the vacuum permittivity; χ_{ii}^Ω is linear dielectric susceptibility at the frequency $\Omega \equiv \omega$ or 2ω ; $\alpha_{2jkl}^{0,\omega,\omega,2\omega}$ are the components of the fourth order polar tensor being invariant with respect to the symmetry operation of the point group D_{2h} of the paraelectric phase; $P_2(0)$ is the Y -component of static spatially homogeneous polarization; and $P_j(\omega,k_1)$, $P_k(\omega,k_2)$, and $P_l(2\omega,k_3)$ are the traveling polarization waves which can be presented as follows:

$$\begin{aligned} P_j(\omega,k_1) &= P_j(k_1) \exp(i\omega t), \\ P_k(\omega,k_2) &= P_k(k_2) \exp(i\omega t), \\ P_l(\omega,k_3) &= P_l(k_3) \exp(i2\omega t), \end{aligned} \quad (7)$$

with the complex amplitudes given as:

$$\begin{aligned} P_j(k_1) &= |P_j(k_1)| \exp(ik_1 r), \\ P_k(k_2) &= |P_k(k_2)| \exp(ik_2 r), \\ P_l(k_3) &= |P_l(k_3)| \exp(ik_3 r). \end{aligned} \quad (8)$$

The coupling α -term in fact describes the NLO conversion process, i.e., the interaction between the homogeneous polarization $P_2(0)$, spatially inhomogeneous polarizations $P_j(\omega,k_1)$ and $P_k(\omega,k_2)$, induced by incident laser light and oscillating at the fundamental frequency ω , and its second temporal harmonics $P_l(2\omega,k_3)$. One may realize that the temporal variable in Eq. (6) has been added only formally. Indeed the free energy density appears to be time independent as it follows after inserting Eq. (7) into Eq. (6). One then obtains the following:

$$f_{\text{int}}(r) = -\alpha_{2jkl}^{0,\omega,\omega,2\omega}[P_2(0)P_j(k_1)P_k(k_2)P_l^*(k_3) + c.c.] + \frac{1}{2\varepsilon_0\chi_{ii}^\Omega}P_i(K)P_i^*(K). \quad (9)$$

This expression is similar to the one used in Ref. 5. The difference is that the Eq. (1) of Ref. 5 is written for the particular case when $k_1=k_2=k_3/2$, i.e., the birefringent phase matching between the interacting modes has been assumed there from the beginning. The free energy density Eq. (9) gives therefore a more general description of the problem. Inserting Eq. (8) into Eq. (9) leads to

$$f_{\text{int}}(r) = -2\alpha_{2jkl}^{0,\omega,\omega,2\omega}P_2(0)|P_j(k_1)||P_k(k_2)||P_l(k_3)|\cos(\Delta kr) + \frac{1}{2\varepsilon_0\chi_{ii}^\Omega}|P_i(K)|^2, \quad (10)$$

where $\Delta k=k_1+k_2-k_3$ is the mismatch wavenumber. The free energy of the NLO interaction corresponding to the active

path length L of the crystal may be obtained by spatial averaging of Eq. (10) which yields:

$$\begin{aligned} F_{\text{int}} &= \frac{1}{L} \int_{-L/2}^{L/2} f_{\text{int}}(r) dr = -2\alpha_{2jkl}^{0,\omega,\omega,2\omega}P_2(0)|P_j(k_1)| \\ &\times |P_k(k_2)||P_l(k_3)|\text{sinc}(\Delta kL/2) + \frac{1}{2\varepsilon_0\chi_{ii}^\Omega}|P_i(K)|^2. \end{aligned} \quad (11)$$

It must be emphasized that the free energy Eq. (11) is indeed obtained ignoring the spatial variation in the real amplitudes $|P_j(k_1)|$ and $|P_k(k_2)|$ which returns us to a local approximation. The amplitude of the polarization wave with doubled frequency $|P_l(k_3)|$ can be derived by minimization of the free energy ($\partial F_{\text{int}}/\partial|P_l(k_3)|=0$), which leads to:

$$|P_l(k_3)| = 2\alpha_{2jkl}^{0,\omega,\omega,2\omega}\varepsilon_0\chi_{ii}^{2\omega}P_2(0)|P_j(k_1)| \times |P_k(k_2)|\text{sinc}(\Delta kL/2). \quad (12)$$

If the phase matching may be achieved ($\Delta k=0$), which indeed is the case of $\text{AgNa}(\text{NO}_2)_2$ crystals, the sinc-factor in Eq. (11) equals 1. This equation then coincides with the one given in Ref. 5. Using the macroscopic relations $|P_j(k_1)|=\varepsilon_0\chi_{jj}^\omega|E_j(\omega)|$ and $|P_k(k_2)|=\varepsilon_0\chi_{kk}^\omega|E_k(\omega)|$ one gets the second order NLO susceptibility as follows:

$$\chi_{ljk}^{(2)} = 2\varepsilon_0^2\alpha_{2jkl}^{0,\omega,\omega,2\omega}P_2(0)\chi_{ll}^{2\omega}\chi_{jj}^\omega\chi_{kk}^\omega. \quad (13)$$

This equation is indeed equivalent to the Eq. (4) of Ref. 15 taking into account that its third order tensor components $\delta_{jkl} \equiv 2\alpha_{2jkl}^{0,\omega,\omega,2\omega}P_2(0)$. Above T_c and at $E=0$ the static polarization $P_2(0) \equiv P_s = 0$ thus all the $\delta_{jkl}=0$ as required by centrosymmetry of the paraelectric phase.

If $E=0$ then $P_2(0) \equiv P_s$. In the paraelectric phase ($T > T_c$) $P_s=0$ thereby all the components $\chi_{ljk}^{(2)}=0$, and the SHG is absent. In the ferroelectric phase ($T \leq T_c$) $\chi_{ljk}^{(2)} \propto P_s$. Both $\chi_{ljk}^{(2)}$ and P_s are expected to exhibit an identical temperature behavior [see solid blue curve in Fig. 1(a)], i.e., the SHG is driven here by spontaneous polarization only.

In the case when an external electric field is applied along the polar axis ($E \parallel Y$) $P_2(0) \equiv P^+$. Accordingly, the SHG is expected to be observed both above and below T_c whereas $\chi_{ljk}^{(2)} \propto P^+$ as it is shown schematically in the Fig. 1(a) [see dashed red curve]. Especially interesting is the temperature range above T_c . Here P^+ is defined exclusively by the electric field induced polarization P_{ind} having anomalous increase in the vicinity of T_c . Since $P_{\text{ind}}=\varepsilon_0(\varepsilon-1)E$ the NLO constant β , defined as $d\chi_{ljk}^{(2)}/dE$, may be expressed as follows:

$$\beta = 2\varepsilon_0^3(\varepsilon-1)\alpha_{2jkl}^{0,\omega,\omega,2\omega}\chi_{ll}^{2\omega}\chi_{jj}^\omega\chi_{kk}^\omega. \quad (14)$$

In the proper ferroelectrics $\varepsilon \gg 1$, thus the temperature variations $\beta(T)$ and $\varepsilon(T)$ are expected to be identical. In the vicinity of T_c they anomalously rise as schematically shown in Fig. 1(b).

III. EXPERIMENTAL

Single crystals of $\text{AgNa}(\text{NO}_2)_2$ were grown from aqueous solution containing 9.8 wt % AgNO_2 and 37.2 wt % NaNO_2 by the slow evaporation method at constant tempera-

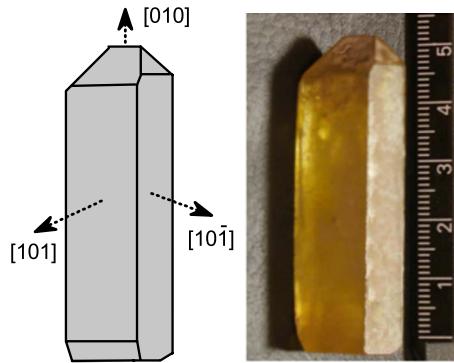


FIG. 2. (Color online) Habit of $\text{AgNa}(\text{NO}_2)_2$ crystal and its crystallographic orientation.

ture ($\sim 25^\circ\text{C}$). Figure 2 shows the habit of the grown single crystals and its crystallographic orientation. We used the standard crystallographic axes set for the paraelectric phase; $a=8.05\text{ \AA}$, $b=10.77\text{ \AA}$, and $c=10.76\text{ \AA}$.²⁰ Crystals $\text{AgNa}(\text{NO}_2)_2$ are yellowish with the perfect cleavage plane perpendicular to $[101]$ and $[10\bar{1}]$ -directions. Accordingly, the geometry of samples was adjusted to these planes, i.e., we used the slabs or plates having perfectly cleaved faces $[101]$ (or $[10\bar{1}]$)⁷ without their further polishing. The silver paste electrodes were deposited on conventionally polished $[010]$ -faces.

The SHG was excited by means of nanosecond IR Q-switched laser Vector 1064-3000-30 ($\lambda=1064\text{ nm}$) and registered in a standard setup described earlier in Ref. 21. The determination of the NLO coefficients associated with SHG was performed by the Maker fringe technique²² as described in detail by Jerphagnon and Kurtz.²³ The halfwave retarder and Glan-Thomson polarizer was used to set polarization (horizontal or vertical) of the incident laser beam depending on phase matching geometry (PMG) being chosen in the experiment. The samples were set into thermostabilized optical cell operating by a temperature controller with an accuracy of the temperature stabilization of about 0.01°C . The thermostabilized cell was attached to an optical table in which the sample has been rotated by means of a step motor with angular step of 0.01° and rotation speed $1\text{--}5\text{ deg/min}$. The outgoing SHG light was filtered by means of interference filter(s) ($\lambda=532\text{ nm}$, $\Delta\lambda=4\text{ nm}$) and registered by photomultiplier. Consequently, the signal was detected using a box car integrator and then recorded.

IV. RESULTS AND DISCUSSION

In the paraelectric phase of $\text{AgNa}(\text{NO}_2)_2$ crystals the SHG is symmetry forbidden. In the ferroelectric phase it is allowed by the symmetry, however these crystals exhibit only a very weak intense SHG signal if the light propagates along the $[101]$ or the symmetry equivalent $[10\bar{1}]$ -direction. We explain this by a lack of the phase matching in NLO interaction along these crystallographic directions. Nevertheless, the phase matching indeed can be achieved in slightly tilted geometries. The insets in Fig. 3 demonstrate the two examples of such PMGs called hereafter as PMG-I and PMG-II, respectively. The PMG-I is defined by two angles

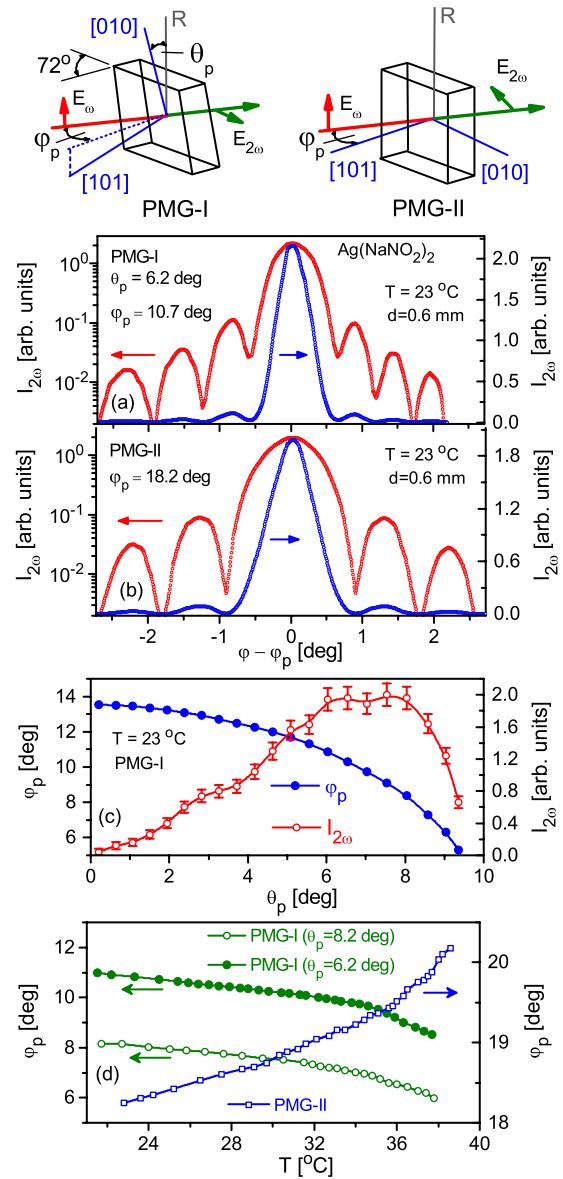


FIG. 3. (Color online) Phase matching in $\text{AgNa}(\text{NO}_2)_2$ crystals. The insets show two examples of PMGs (PMG-I and PMG-II). Panels (a) and (b) show the angular dependencies of the SHG intensity $I_{2\omega}$ in the vicinity of the phase matching angle φ_p at $T=23^\circ\text{C}$ for PMG-I and PMG-II, respectively. (c) θ_p -dependence of φ_p and $I_{2\omega}$ in PMG-I at $T=23^\circ\text{C}$. (d) Temperature dependences of φ_p at fixed and θ_p in PMG-I and PMG-II.

θ_p and φ_p . The incident intense nanosecond-pulsed light ($\lambda=1064\text{ nm}$, $\tau=300\text{ ps}$) is polarized vertically whereas the outgoing SHG light ($\lambda=532\text{ nm}$) exhibits nearly horizontal polarization at small θ_p . The angle φ_p depends on θ_p as shown in Fig. 3(c) whereas SHG intensity $I_{2\omega}$ passes through its maximum magnitudes at $\theta_p \approx 6^\circ\text{--}8^\circ$. For the PMG-II the rotation R -axis is set perpendicularly to the $[101]$ and $[010]$ -directions. The phase matching direction is defined here by symmetry equivalent angles $\pm\varphi_p(\theta_p=0)$, incident laser beam is polarized vertically whereas outgoing intense SHG exhibits its polarization turned on of angle about $\pm 52^\circ$ ($T=23^\circ\text{C}$). In both geometries, the φ -dependence of SHG intensity $I_{2\omega}$ reveals Maker-fringe pattern [see Figs. 3(a) and 3(b)] which in in a close vicinity of φ_p roughly scales with $\propto L^2(\chi_{\text{eff}}^{(2)})^2 \text{sinc}^2[a(\varphi - \varphi_p)L/2]$, where L and $\chi_{\text{eff}}^{(2)}$ are the path

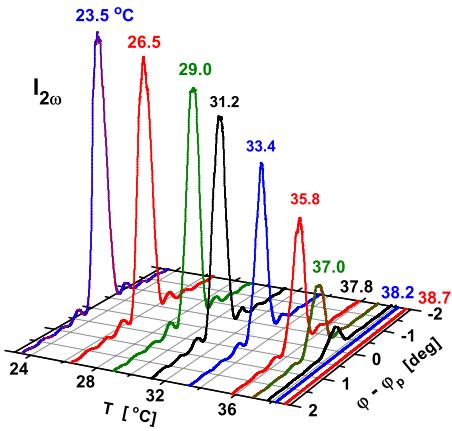


FIG. 4. (Color online) The angular φ -dependences of the SHG intensity $I_{2\omega}$ in the vicinity of the phase matching angle $\varphi_p(\theta=\theta_p)$ at different temperatures. The maximum of $I_{2\omega}$ corresponds to PMG-I defined by the angles φ_p and θ_p .

length of NLO interaction and the effective second order NLO susceptibility, respectively, coefficient a define the angular dependence of the mismatch vector Δk in linear approximation [$\Delta k=a(\varphi-\varphi_p)$]. A challenging question is a type of the observed phase matching, i.e., whether it is of birefringent origin or one deals with so-called quasiphase matching caused by eventual ferroelectric domain structure. An exact answer on this question could be obtained having measured the principal refractive indices at $\lambda=532$ and 1064 nm. Unfortunately, the refractometry measurements appear to be really problematic along the principal crystallographic directions of $\text{AgNa}(\text{NO}_2)_2$ crystals. Due to a perfect splitting of the single crystals (parallel to $[101]$ and/or $[10\bar{1}]$ cleavage planes) we were not able to prepare crystal cuts of sufficiently good optical quality along other crystallographic directions, in particular, the cuts perpendicular to the principal crystallographic directions which are required for eventual interferometric and/or refractometric measurements. On the other hand, it must be emphasized that in our case the measurements were performed on fresh samples being cut from the crystals that were grown in the ferroelectric phase. The obtained in such way crystals are usually single domain. The results were reproduced on several samples. Also the electric field up to 3 kV/cm applied along the polar axis did not change the SHG signal. For these reasons we believe that one deals with the birefringent phase matching rather than with the quasiphase one provided by a regular ferroelectric domain structure.

Figure 4 shows the angular dependences of the SHG intensity $I_{2\omega}$ of $\text{AgNa}(\text{NO}_2)_2$ crystals in the vicinity of the phase matching angle φ_p being recorded at different temperatures. The maximum of $I_{2\omega}(\varphi)$ corresponds here to PMG-I defined by angles φ_p and θ_p . One can see that the character of the Maker fringe patterns itself does not change significantly while the temperature rises. Only the SHG signal significantly decreases by approaching the ferroelectric-to-paraelectric PT and completely vanishes at $T\geq T_c$.

In accordance with the point group C_{2v} the effective susceptibility $\chi_{\text{eff}}^{(2)}$ in ferroelectric phase is expressed through the partial contributions of NLO tensor components $\chi_{222}^{(2)}$, $\chi_{233}^{(2)}$,

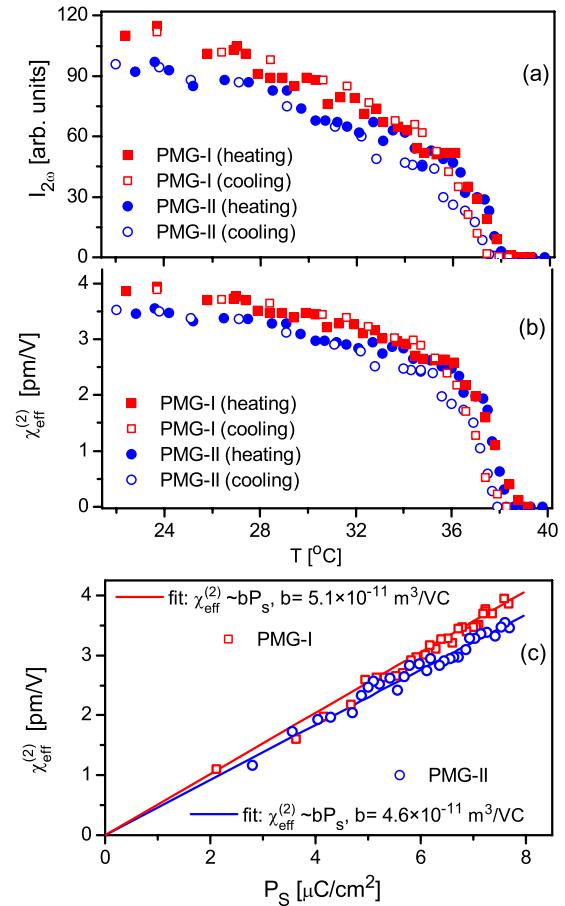


FIG. 5. (Color online) SHG in ferroelectric $\text{AgNa}(\text{NO}_2)_2$ crystals. (a) Temperature dependences of the SHG intensity $I_{2\omega}$ measured in PMG-I and PMG-II at heating and cooling. (b) Effective NLO susceptibility $\chi_{\text{eff}}^{(2)}$ vs T determined in the reference measurements $[I_{2\omega}(T)/I_{2\omega}^{\text{SiO}_2}(\chi_{111})]$ at heating and cooling in PMG-I (solid and open squares) and PMG-II (solid and open circles), respectively. (c) $\chi_{\text{eff}}^{(2)}$ vs the spontaneous polarization P_s in PMG-I and PMG-II at heating [$P_s(T)$ -dependence has been taken from Ref. 8]. Solid lines are the best linear fits of $\chi_{\text{eff}}^{(2)}(P_s)$. Their slopes define the magnitudes of the coupling constant b .

$\chi_{211}^{(2)}$, $\chi_{121}^{(2)}$, and $\chi_{323}^{(2)}$ each of which is proportional to the spontaneous polarization P_s as has been shown in Sec. II. On the other hand, the path length L and the effective susceptibility $\chi_{\text{eff}}^{(2)}$ are both in principle dependent on the angles θ and φ . However at fixed $\theta=\theta_p$ the other phase matching angle φ_p , corresponding to the maximum of SHG intensity, is only very weakly temperature dependent in the ferroelectric phase for both PMGs [see Fig. 3(d)]. Basing on these facts our analysis shows that the weight factors bonded to the partial contributions of $\chi_{ijk}^{(2)}$ into $\chi_{\text{eff}}^{(2)}$, either do not depend at all or change only very slightly with the temperature due to a weakly varying $\varphi_p(T)$ -dependence. Accordingly, $\chi_{\text{eff}}^{(2)}(T)$ in the ferroelectric phase scales with the temperature dependence of the square root of $I_{2\omega}$ measured at both PMGs and is expected to be proportional to the spontaneous polarization P_s . This conclusion will be subjected to verification, see discussion just below.

Figure 5(a) shows the temperature dependences of $I_{2\omega}$ determined at heating and cooling for PMG-I (solid and open circles) and PMG-II (solid and open squares), respectively. In the following we compare it with the intensity of SHG in

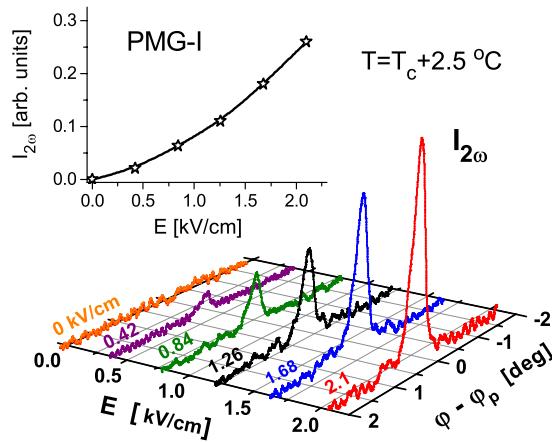


FIG. 6. (Color online) The angular dependences of the SHG intensity $I_{2\omega}$ in the vicinity of the phase matching angle φ_p for PMG-I measured at different applied electric fields in the paraelectric phase ($T=T_c+2.5$ °C). Inset shows the electric field dependence of $I_{2\omega}$ being extracted at each E as maxima of the presented angular φ -dependences in the figure.

SiO_2 ($\chi_{111}^{(2)}$ -component), one obtains at $T=23$ °C the magnitudes $\chi_{\text{eff}}^{(2)}(\text{PMG-I})=4.0$ pm/V and $\chi_{\text{eff}}^{(2)}(\text{PMG-II})=3.6$ pm/V. Figure 5(b) shows the temperature dependences of $\chi_{\text{eff}}^{(2)}$ determined in PMG-I and PMG-II. The ferroelectric-to-paraelectric PT is accompanied by a vanishing of SHG, however the variations in $\chi_{\text{eff}}^{(2)}(T)$ in the vicinity of the Curie point appears to be smeared over the temperature range of about 1.5 °C and are characterized for both PMGs by the temperature hysteresis of about 0.5 °C. A smeared character of this PT may be an evidence for coexistence of metastable polar and nonpolar regions indicating on a sufficiently spatially inhomogeneous structure that occurs in the region of the Curie point T_c . Using the spontaneous polarization $P_s(T)$ measured in Ref. 8 we present in Fig. 5(c) the dependences of $\chi_{\text{eff}}^{(2)}$ versus P_s . They appear to be linear for both PMGs; $\chi_{\text{eff}}^{(2)}=bP_s$, where b is the characteristic effective constant. Linear fitting of the $\chi_{\text{eff}}^{(2)}(P_s)$ -dependence allows to derive directly the magnitude of the constant b which appears to be equal to 5.1×10^{-11} and 4.6×10^{-11} m³/V C as for PMG-I and PMG-II, respectively. Within such approach the quadratic nonlinearity is considered to be induced by the spontaneous polarization, thus we deal with the spontaneous SHG as has been mentioned above. In the paraelectric phase the SHG does not occur since $P_s=0$. On the other hand, this behavior can be also interpreted applying the symmetry principle. According to it the quadratic nonlinearity does not occur in the paraelectric phase because in the centrosymmetric media all the tensor components $\chi_{njjm}^{(2)}$ are equal 0. The spontaneous polarization P_s , that appears below T_c , plays here a role of a symmetry breaking parameter which lowers the symmetry from the centrosymmetric D_{2h} to the noncentrosymmetric C_{2v} group.

Following this route, the symmetry can be lowered as well due to the induced polarization P_{ind} caused by an applied external field E . In this case $\chi_{\text{eff}}^{(2)}=bP_{\text{ind}}$, where $P_{\text{ind}}=\epsilon_0(\epsilon-1)E$, ϵ is the static dielectric constant. Figure 6 reports the SHG in the centrosymmetric paraelectric phase of $\text{AgNa}(\text{NO}_2)_2$ driven by the electric field being applied along the Y-axis. Inset of Fig. 6 demonstrate the field dependence

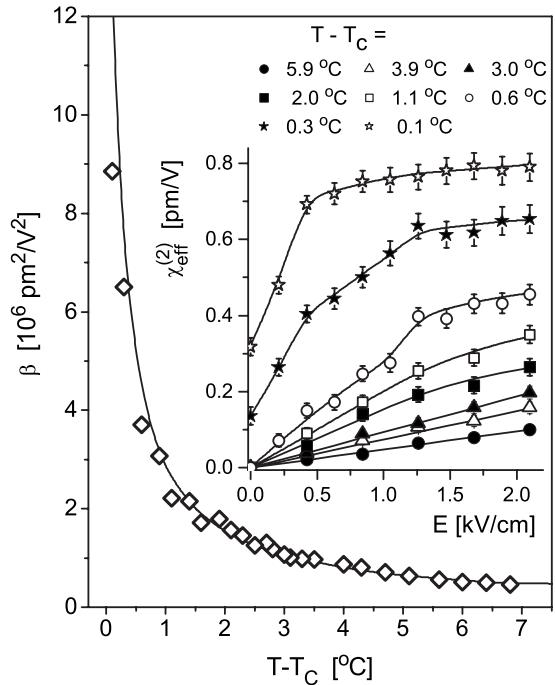


FIG. 7. Temperature dependence of the NLO coefficient $\beta=d\chi_{\text{eff}}^{(2)}/dE$ determined in PMG-I in the paraelectric phase. β has been obtained in the region of small electric field ($E\rightarrow 0$) being applied along the polar Y-axis. Inset shows $\chi_{\text{eff}}^{(2)}(E)$ -dependences measured at several temperatures above T_c .

of the $I_{2\omega}$ being extracted as maxima of φ -dependences shown in Fig. 6, i.e., at PMG-I geometry ($\varphi=\varphi_p$, $\theta=\theta_p$). Square root of SHG signal ($I_{2\omega}^{1/2}$) is again proportional to $\chi_{\text{eff}}^{(2)}$. Far above T_c the $\chi_{\text{eff}}^{(2)}(E)$ -dependences are nearly linear, but approaching T_c they gradually becomes more nonlinear, see inset in Fig. 7. In the close vicinity of PT-point the $\chi_{\text{eff}}^{(2)}(E)$ -dependences get a saturated character reproducing the shape of $P_{\text{ind}}(E)$ -dependences being typical for ferroelectric materials in this temperature range.⁶ The constant β , defining as $d\chi_{\text{eff}}^{(2)}/dE$, describes a cubic nonlinearity and is $\propto dP_{\text{ind}}/dE \propto \epsilon$. At small electric field the T -dependence of dielectric constant $\epsilon(T)$ follows the Curie-Weiss law $C_0/(T-T_0)$,⁷ thus one can expect similar anomalous rising of the NLO constant β in the vicinity of ferroelectric Curie point T_c . Figure 7 demonstrates the temperature dependence of β , determined from the slopes of $\chi_{\text{eff}}^{(2)}(E)$ -dependences at $E\rightarrow 0$. One can realize that $\beta(T)$ behaves in analogical manner as static dielectric constant $\epsilon(T)$, i.e., anomalously rises by approaching the Curie point T_c . The $\epsilon(T)$ -dependence can be indeed derived from the $\beta(T)$ -dependence assuming that one deals with the small applied fields E . On the one hand, this means that $\chi_{\text{eff}}^{(2)}=\beta E$. On the other hand, the second order NLO susceptibility $\chi_{\text{eff}}^{(2)}$ can be expressed through the electric field induced polarization P_{ind} , i.e., $\chi_{\text{eff}}^{(2)}=bP_{\text{ind}}=b\epsilon_0(\epsilon-1)E$. By comparing these expressions one obtains quite simple relation as follows:

$$\epsilon = 1 + \beta/(b\epsilon_0). \quad (15)$$

Figure 8 shows the static dielectric constant ϵ and its inverse magnitude ϵ^{-1} versus T being determined in the paraelectric phase by using the $\beta(T)$ -dependence, given in Fig. 7, and coupling constant $b=5.1\times 10^{-11}$ m³/C V, as derived from

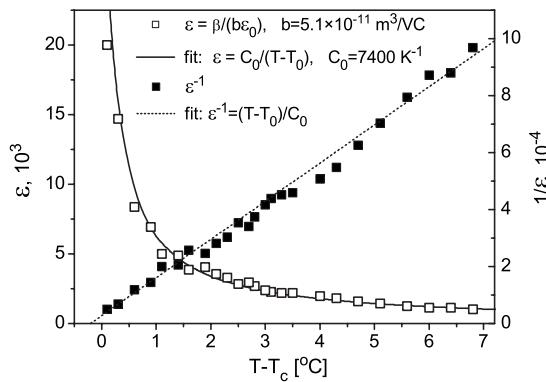


FIG. 8. Static dielectric constant ϵ and its inverse magnitude ϵ^{-1} vs T being determined in the paraelectric phase from $\beta(T)$ -dependence [$\epsilon(T) = \beta(T)/(\epsilon_0 b)$]. Solid lines are the best Curie-Weiss fits providing the magnitude of the Curie-Weiss constant $C_0 = 7400 \text{ K}^{-1}$.

$\chi_{\text{eff}}^{(2)}(P_s)$ -dependence [see Fig. 5(c)]. Solid lines are the best Curie-Weiss fits providing the magnitude of the Curie-Weiss constant $C_0 = 7400 \text{ K}^{-1}$ for $\text{AgNa}(\text{NO}_2)_2$ crystal. This value is larger comparing to the one ($C_0 = 4700 \text{ K}^{-1}$) given in Ref. 7. However, it is very likely that the C_0 value reported in that work has not been correctly extracted from the presented there $\epsilon^{-1}(T)$ -dependence. Our rough estimation of the data⁷ gives somewhat larger its magnitude, i.e., $C_0 \approx 5900 \text{ K}^{-1}$. Moreover, by comparing the C_0 magnitudes obtained by different techniques one must keep in mind different experimental conditions under which the measurements have been performed. In particular, our NLO measurements correspond to a fully static regime ($\omega/2\pi=0$) whereas the conditions in Ref. 7 are quasistatic ones ($\omega/2\pi=1000 \text{ Hz}$). This remark appears to be crucial especially for $\text{AgNa}(\text{NO}_2)_2$ crystals which are characterized by extremely slow polarization dynamics corresponding to the kilohertz region.^{9,13,14} Due to this reason the difference $\Delta T = T_c - T_0 \approx 2 \text{ }^\circ\text{C}$ obtained in (Ref. 7) appears to be substantially larger comparing to the one derived in the present work ($\Delta T \approx 0.25 \text{ }^\circ\text{C}$) by extrapolating linear dependence $\epsilon^{-1}(T)$ into the region below T_c (see Fig. 8). Really small magnitude of ΔT can be considered here as an evidence that we indeed deal with the PT of the first order being very close to the tricritical point.¹⁰

V. CONCLUSION

Taken together, we report here on SHG in ferroelectric $\text{AgNa}(\text{NO}_2)_2$ crystals being driven by the spontaneous and electric field induced polarizations. Obtained results are interpreted within the phenomenological theory which considers the free energy describing the interaction between the spontaneous or electric field induced polarizations and spatially inhomogeneous electric polarizations resulted from propagating optical waves. At room temperature $\text{AgNa}(\text{NO}_2)_2$ crystals are characterized by the effective NLO susceptibilities $\chi_{\text{eff}}^{(2)}$ being comparable or significantly larger comparing to a large number of other known NLO materials such as inorganic crystals SiO_2 ,²⁴ KH_2PO_4 ,²⁴ PbHPO_4 ,²⁵ $\text{BaAlBO}_3\text{F}_2$,²⁶ $\text{Bi}_2\text{ZnOB}_2\text{O}_6$ (Ref. 27) or organic crystals 3- and 4-aminophenol- H_3PO_4 ,²⁸ L-histidinium-L-tartrate hemihydrate²⁹ and $(\text{H}_3\text{N}(\text{CH}_2)_2\text{SS}(\text{CH}_2)_2\text{NH}_3^+$

$\text{PbI}_5\cdot\text{H}_2\text{O}$.^{30,31} However, it is more important that relatively high magnitudes of the effective NLO susceptibilities are combined in $\text{AgNa}(\text{NO}_2)_2$ crystals with several PMGs. This allows to consider them as high-performance materials for potential NLO applications, such as frequency doubling, parametric generation and amplification, or other applications that require a high-efficient frequency conversion. In addition, an anomalously large response of NLO susceptibilities with respect to an applied electric field has been found in the vicinity of the Curie point of $\text{AgNa}(\text{NO}_2)_2$ crystals. This may also have a number of applications, especially in those devices where an efficient tunable control of SHG intensity is demanded.

ACKNOWLEDGMENTS

This work has been supported by grant Moltech-Anjou (Angers, France) likewise by the Program of the Ukrainian-Polish Scientific-Technical Cooperation in Years 2009–2010 [Project title: “Investigations of new crystalline materials for optoelectronics applications: characterization, geometry optimization, and improvement of their efficiency in practical applications” Project No. M/138-2009 (according to Ukrainian classification); Project No.: 16 (according to Polish classification)].

- ¹B. E. A. Saleh and M. C. Teich, *Fundamentals of Photonics* (Wiley, New York, 1991).
- ²F. Agulló-López, J. M. Cabrera, and F. Agulló-Rueda, *Electro-Optics, Phenomena, Materials, and Applications* (Academic, New York, 1994).
- ³R. Dhanasekaran, *IOP Conf. Ser.: Mater. Sci. Eng.* **2**, 012014 (2009).
- ⁴M. DrDomenico, Jr. and S. H. Wemple, *J. Appl. Phys.* **40**, 720 (1969).
- ⁵A. V. Kityk, R. Czaplicki, A. Klöpperpieper, A. S. Andrushchak, and B. Sahraoui, *Appl. Phys. Lett.* **96**, 061911 (2010).
- ⁶F. Jona and G. Shirane, *Ferroelectric Crystals* (Dover, New York, 1993).
- ⁷K. Gesi, *J. Phys. Soc. Jpn.* **28**, 395 (1970).
- ⁸K. Gesi, *J. Phys. Soc. Jpn.* **33**, 108 (1972).
- ⁹J. Petersson, E. Schneider, and R. Siems, *Z. Phys. B: Condens. Matter* **39**, 233 (1980).
- ¹⁰J. Helwig, J. Petersson, and E. Schneider, *Z. Phys. B* **28**, 87 (1977).
- ¹¹V. P. Sopranyuk, A. Fuith, H. Kabelka, K. Knorr, A. Klöpperpieper, K. Sokalski, and A. V. Kityk, *Phys. Rev. B* **66**, 104102 (2002).
- ¹²J. Grossmann, D. Muller, and J. Petersson, *Z. Naturforsch. A* **29A**, 1055 (1974).
- ¹³J. Grossmann, D. Muller, J. Petersson, and E. Schneider, *Z. Naturforsch. A* **31A**, 1089 (1976).
- ¹⁴Y. Tominaga, S. Wada, and S. Iida, *J. Phys. Soc. Jpn.* **32**, 1675 (1972).
- ¹⁵R. C. Miller, *Appl. Phys. Lett.* **5**, 17 (1964).
- ¹⁶A. V. Kityk, O. M. Mokry, V. P. Sopranyuk, and O. G. Vlokh, *J. Phys.: Condens. Matter* **5**, 5189 (1993).
- ¹⁷K. T. Mashiyama and H. Mashiyama, *J. Phys. Soc. Jpn.* **56**, 1810 (1987).
- ¹⁸O. G. Vlokh, A. V. Kityk, O. M. Mokry, and V. G. Grybyk, *Phys. Status Solidi A* **116**, 287 (1989).
- ¹⁹A. V. Kityk, V. P. Sopranyuk, A. Fuith, W. Schranz, and H. Warhanek, *Phys. Rev. B* **53**, 6337 (1996).
- ²⁰H. Miki, Y. Makita, and K. Gesi, *J. Phys. Soc. Jpn.* **30**, 1512 (1971).
- ²¹B. Sahraoui, J. Luc, A. Meghea, R. Czaplicki, J.-L. Fillaut, and A. Migalska-Zalas, *J. Opt. A, Pure Appl. Opt.* **11**, 024005 (2009).
- ²²P. D. Maker, R. W. Terhue, M. Nisenoff, and C. M. Savage, *Phys. Rev. Lett.* **8**, 21 (1962).
- ²³J. Jerphagnon and S. K. Kurtz, *J. Appl. Phys.* **41**, 1667 (1970).
- ²⁴I. Shoji, T. Kondo, A. Kitamoto, M. Shirane, and R. Ito, *J. Opt. Soc. Am. B* **14**, 2268 (1997).
- ²⁵A. Keens and H. Happ, *J. Phys. C* **21**, 1661 (1988).
- ²⁶Z.-G. Hu, M. Yoshimura, K. Muramatsu, Y. Mori, and T. Sasaki, *Jpn. J. Appl. Phys., Part 2* **41**, L1131 (2002).
- ²⁷F. Li, S. Pan, X. Hou, and J. Yao, *Cryst. Growth Des.* **9**, 4092 (2009).

- ²⁸T. Głowiąk, S. Debrus, M. May, A. J. Barnes, and H. Ratajczak, *J. Mol. Struct.* **596**, 77 (2001).
- ²⁹M. K. Marchewka, S. Debrus, A. Pietraszko, A. J. Barnesc, and H. Ratajczak, *J. Mol. Struct.* **656**, 265 (2003).

- ³⁰W. Bi, N. Louvain, N. Mercier, J. Luc, I. Rau, F. Kajzar, and B. Sahraoui, *Adv. Mater. (Weinheim, Ger.)* **20**, 1013 (2008).
- ³¹N. Louvain, N. Mercier, J. Luc, and B. Sahraoui, *Eur. J. Inorg. Chem.* **2008**, 3592.